## **APPENDICES**

Appendix A: The chemical composition of austenitic stainless steels developed for biomedical applications (mass %).

Designation	C	Mn	Cr	Ni	Mo	N	Si	Р	S	Cu	Nb	Fe
F 138 [211]	0.03	2.0	17.0-	13.0-	2.25-	0.1	0.75	0.025	0.01	0.50	-	Bal
	max	max	19.0	15.0	3.0	max	max	max	max	max		
ASTM F1314 [89]	0.03	4.0-6.0	20.5-	11.5-	2.0-	0.2-	0.75	0.025	0.01	0.5	0.1-0.3	Bal
	max		23.5	13.5	3.0	0.4	max	max	max	max		
ASTM F1586 [90]	0.08	2.0-	19.5-	9.0-	2.0-	0.25-	0.75	0.025	0.01	0.25	0.25-	Bal
	max.	4.25	22.5	11.0	3.0	0.50	max.	max.	max.	max.	0.80	
ASTM F2581 [95]	0.15-	9.5-	16.5-	0.05	2.7-	0.45-	0.2-	0.02	0.01	0.25	-	Bal
	0.25	12.5	18.0	max	3.7	0.55	0.6	max.	max	max		
ASTM F2229 [94]/ Biodur	0.08	21.0-	19.0-	0.05	0.5-	0.85-	0.75	0.03	0.01	0.25	-	Bal
108 [68]	max	24.0	23.0	max	1.5	1.1	max	max	max	max		
P2000/ X13CrMnMoN18-	< 0.15	12-16	16-20	-	2.5-	0.75-	-	-	-	-	< 0.25	Bal
14-3 [115]					4.2	1.0						
PANACEA P558 [118]	0.16	9.47	16.62	< 0.02	3.25	0.49	0.43	< 0.005	0.0002	0.02	0.026	Bal
BIOSSN4 [212]	0.029	12.58	17.05	0.03	2.38	0.43	0.42	0.014	0.007	1.44	-	Bal
0Cr18Mn15Mo2N0.64	0.044	15.8	18.62	0.03	2.78	0.64	0.18	0.013	0.004	-	-	Bal
[125]												

Appendix B: Microstructure and mechanical behavior of nickel-free/negligible nickel, nitrogen stabilized austenitic stainless steel.

S. No.	Alloys/Composition	Experimental details	Findings	References
1.	Cr-19, Mn-10, N-0.63,	The tribological behavior was	In a corrosive medium, Cr-Mn-N steel	1998 [97]
	Ni-0.8, Mo-0.16, C-	investigated by dry abrasion, abrasion-	exhibited superior abrasion resistance and	
	0.03, Si-0.03	corrosion (synthetic mine water) and	cavitation-erosion resistance.	
		cavitation-erosion (in distilled water)		
		techniques and compared with that of		
		the 304 and Hadfield steel.		
2.	P2000	Fatigue, chemical and sliding wear in	It showed good corrosion, fatigue and	2002
		Ringer's solution were studied. In vitro	tribological behavior and is recommended for	[115,116]
		cytotoxicity was studied with MC3T3	biomedical application.	
		cells.		
3.	Biodur 108, 316L and	Fatigue tests were conducted in	The S-N curve of Biodur 108 is significantly	2003 [98]
	22Cr-13Ni-5Mn	Ringer's solution at 37°C as per ASTM	above the 316L and fatigue life decreased in	
		F1801 up to failure or $10^6$ cycles at 1 Hz	Ringer's solution compared to distilled water	
		frequency.	for all three steels.	
4.	Fe24Mn13Cr1Ni0.44N,	Tests under tensile loading were	Slip and twinning both the mechanisms were	2006 [99]
	Fe24Mn18Cr3Ni0.62N	conducted at an elongation rate of 10	there during the testing.	
		$\mu$ m/min and 50 $\mu$ m/min. The		

	microstructure was analyzed using in-		
	situ SEM.		
Fe-24Cr-2Mo (HFVM),	High cycle fatigue test at stress ratio (R)	No significant difference between S-N curves	2009 [102]
Fe-24Cr-2Mo-1N (NA-	of 0.1in air and PBS solution kept at	was there in air and in PBS for each steel. The	
1) and Fe-23Cr-1Mo-	310°K and pH 7.5 bubbled with	fatigue strength at $10^7$ cycles for the NA alloy	
1N (P-ESR)	nitrogen-4% O <sub>2</sub> gas mixture, with 20	and P-ESR alloy was 245 MPa and 320 MPa,	
	Hz frequency in air and 2 Hz in PBS	respectively.	
	solution.		
Fe24Mn13Cr0.44N	Bend rotating cyclic tests were	The fatigue strength at $10^7$ cycles was 341	2009 [101]
	performed at room temperature at 40 Hz	MPa.	
	frequency and R= -1		
ASTM F2229 and 316L	Fatigue tests were performed at RT with	Cracks were preferentially initiated along	2013, 2014
in cold rolled condition	R=0.053 and frequency of 1 Hz. Tests	annealing twin boundaries for ASTM F2229,	[100,210]
(20-30%)	were interrupted to analyze the crack	whereas for 316L, cracks initiated along	
	initiation mechanisms.	intrusions and extrusions associated with slip.	
HNS (C-0.022, Si-0.15,	Fretting fatigue behavior of HNS was	Fatigue strength of HNS at 10 <sup>7</sup> cycles was	2013 [103]
Mn-0.10, P< 0.05, S-	studied in air and PBS kept at pH of 7.5	320 MPa in air and PBS and was comparable	
0.0005, Ni-0.04, Cr-	and temperature of 310 °K and aerated	to that of 316L. The fretting fatigue was 280	
23.78, Mo-0.96, N-	with 4% O <sub>2</sub> containing N <sub>2</sub> gas mixture.	MPa and 240 MPa in air and PBS,	
1.05) and 316L (20%	Tests were conducted at R of 0.1 at 20	respectively. The plain fatigue of HNS was	
cold rolled)	Hz and 2 Hz frequency in air and PBS	lower than that of 316L (350 and 320 MPa)	
	Fe-24Cr-2Mo (HFVM),   Fe-24Cr-2Mo-1N (NA-   1) and Fe-23Cr-1Mo-   1N (P-ESR)   Fe24Mn13Cr0.44N   ASTM F2229 and 316L   in cold rolled condition   (20-30%)   HNS (C-0.022, Si-0.15,   Mn-0.10, P< 0.05, S-   0.0005, Ni-0.04, Cr-   23.78, Mo-0.96, N-   1.05) and 316L (20%   cold rolled)	microstructure was analyzed using insitu SEM.Fe-24Cr-2Mo (HFVM),High cycle fatigue test at stress ratio (R)Fe-24Cr-2Mo-1N (NA-of 0.1in air and PBS solution kept at1) and Fe-23Cr-1Mo-310°K and pH 7.5 bubbled with1N (P-ESR)nitrogen-4% O2 gas mixture, with 20Hz frequency in air and 2 Hz in PBS solution.Nutro PESRFe24Mn13Cr0.44NBend rotating cyclic tests were performed at room temperature at 40 Hz frequency and R= -1ASTM F2229 and 316LFatigue tests were performed at RT with in cold rolled condition (20-30%)HNS (C-0.022, Si-0.15, 0.0005, Ni-0.04, Cr- 23.78, Mo-0.96, N- 1.05) and 316L (20% cold rolled)Frequency in air and PBS kept at pH of 7.5 were conducted at R of 0.1 at 20 Cold rolled)	microstructure was analyzed using in- situ SEM.microstructure was analyzed using in- situ SEM.Fe-24Cr-2Mo (HFVM),High cycle fatigue test at stress ratio (R)No significant difference between S-N curves was there in air and in PBS for each steel. The fatigue strength at 107 cycles for the NA alloy and P-ESR alloy was 245 MPa and 320 MPa, respectively.IN (P-ESR)No rotating cyclic tests were performed at room temperature at 40 Hz frequency and R= -1The fatigue strength at 107 cycles was 341 MPa.ASTM F2229 and 316L in cold rolled condition (20-30%)Fatigue tests were performed at RT with initiation mechanisms.Cracks were preferentially initiated along initrusions and extrusions associated with slip.HNS (C-0.022, Si-0.15, 0.0005, Ni-0.04, Cr- 0.0005, Ni-0.04, Cr-Fretting fatigue behavior of HNS was at and PBS kept at pH of 7.5Fatigue strength of HNS at 107 cycles was 280 annealing Nu har and PBS and was comparable to that of 316L. The fretting fatigue was 280 and 240 MPa in air and PBS, respectively.

		solution, at a contact pressure of 30	whereas, the fretting fatigue limit was	
		MPa	significantly higher than that of 316L (180	
			MPa).	
9.	Cr-17.97, Mn-18, N-	Tests were performed under tensile	Increase in strength and decrease in	2015
	0.63, C-0.056, Al-0.02	loading at strain rates of $10^{-4}$ and $10^{-2}$	elongation was observed with increasing	[104,107]
		/sec. LCF tests were performed at RT at	strain rate. The plastic deformation behavior	
		a constant frequency of 0.5 Hz and	was well fitted with modified Ludwik	
		different strain amplitudes.	relation. Cracks nucleation was at grain	
			boundaries or along slip bands at higher strain	
			rate, whereas at lower strain rate, they	
			initiated preferably at inclusions.	
			There was fall in fatigue life with rise in strain	
			amplitude. Cyclic softening was there from	
			the initial cycle.	
10.	316, 316LVM,	Tensile and hardness tests at RT	There was increase in hardness and strength	2015 [105]
	$316MnN_1$ , $316MnN_2$ ,		with increase in nitrogen content while	
	316MnN <sub>3</sub> ,		decrease in ductility.	
11.	Cr-22, Mn-17, Mo-2.43,	Tensile tests were carried out at RT at	This steel exhibited two-stage strain	2017
	N-0.83, C-0.02, Si-0.27,	different strain rates ranging from 10 <sup>-</sup>	hardening behaviour. There was increase in	[108,109]
	Nb-0.21, P-0.012, S-	<sup>4</sup> /sec to 1/sec in the solution treated, hot	YS and UTS with increase in strain rate but	
	0.004	rolled and cold rolled condition	decrease in elongation.	

12.	Fe-18Cr-10Mn-0.4N		High cycle fatigue tests were conducted	Fatigue limits of 0.4N and 0.6N at 10 <sup>7</sup> cycles	2017 [106]
	and Fe-18Cr-10Mn-		with R=0.1 at 10 Hz frequency	were 425 MPa and 475 MPa, respectively.	
	0.6N			Both the steels have similar UTS, but YS of	
				0.6N (537 MPa) is higher than that of 0.4N	
				(445 Mpa)	

Note: SEM: Scanning electron microscopy; PBS: Phosphate buffer saline; YS: Yield strength; UST: Ultimate tensile strength; RT: room temperature

S. No.	Alloys/Composition		Experime	ntal det	ails	Findings	References
		Medium/soluti	pН	Tem	Methods		
		on		p.			
1.	C-0.2, Cr-17.35, Mn-	PBS, MEM,	-	-	EIS	Two maxima observed in the bode	2005 [111]
	10.18, Mo-3.09, N-	MEM+10%				phase spectra indicate the presence of	
	0.49	FCS,				one inner passive layer whose resistance	
		MEM+10%				increased with immersion time related	
		FCS + L929				to uniform corrosion and one outer	
		cells				porous layer.	
2.	C-0.048, Cr-18.44,	XM H <sub>2</sub> SO <sub>4</sub> +	-	-	Potentiodynamic	Passive film formed in NaCl was stable,	2009 [110]
	Mo-2.23, Mn-15.96,	0.5M NaCl,			polarisation, EIS,	but stability decreased with the addition	
	Si-0.24, P-0.004, S-	$0.5M H_2SO_4 +$			Mott-Schottky	of H <sub>2</sub> SO <sub>4</sub> . Passive film formed on acidic	
	0.017, N-0.66	0.25M Na <sub>2</sub> SO <sub>4</sub>			and XPS	Na <sub>2</sub> SO <sub>4</sub> was superior to acidic NaCl	
3.	HNS A (19.8 Cr,	3.5% NaCl,	-	RT	Potentiodynamic	HNS exhibited higher resistance against	2009 [66]
	18.4 Mn, 0.82 N,	0.5 M NaCl +			polarisation, EIS	pitting than that of the 316L in both	
	0.04 C, 0.012 S,	0.5 M H <sub>2</sub> SO <sub>4</sub>				mediums. There was no significant	
	0.015 P, 0.02 Al ),					effect of the addition of Mo.	
	HNS B (19.07 Cr,						
	18.84 Mn, 2.20 Mo,						

Appendix C: Electrochemical corrosion study of nickel-free/negligible nickel, nitrogen stabilized austenitic stainless steel.

	0.77 N, 0.043 C,						
	0.012 S, 0.015 P,						
	0.02 Al ), HNS C						
	(18.4 Cr, 15.8 Mn,						
	2.19 Mo, 0.66 N,						
	0.24 Si, 0.04 C, 0.005						
	S, 0.017 P, 0.02 Al )						
	and 316L						
4.	A1 (19.56 Cr, 19.4	3.5 % NaCl	1, 3, 6,	30	XPS,	Nitrogen steels had better resistance	2009 [84]
	Mn, 2.29 Mo, 0.82 N,	with addition	7 and 9	°C -	Potentiodynamic	against pitting and crevice corrosion	
	0.049 C, 0.19 Si,	of HCl and		70	polarisation and	than that of 316L. Increase in pitting	
	0.003 S, 0.04 Al ), A2	NaOH		°C	EIS	resistance and CPT was observed with	
	(19.84 Cr, 18.9 Mn,					increase in nitrogen content. A relation	
	2.26 Mo, 0.88 N,					between CPT and MARC is given as :	
	0.022 C, 0.19 Si,					CPT= 2.55 MARC – 29	
	0.002 S, 0.02 Al), A3						
	(19.55 Cr, 19.5 Mn,						
	2.26 Mo, 0.96 N,						
	0.058 C, 0.19 Si,						
	0.003 S, 0.04 Al) and						
	316L						

5.	00Cr18Mn15Mo2N0	Hank's	-	37	Potentiodynamic	Better corrosion resistance and blood	2009 [123]
	.62	solution		°C	polarisation,	compatibility as compared to 316L. YS	
					platelet adhesion	and UTS of 537 MPa and 884 MPa,	
					and tensile test	respectively was observed.	
6.	P558 and ISO 5832-9	MEM	-	37	Potentiodynamic	Electrochemical corrosion resistance	2010 [117]
				°C	polarisation and	was comparable to that of the ISO 5832-	
					EIS	9	
7.	Fe-17Cr-15Mn-	PBS, PBS with	7.4	37	EIS, cyclic	HNS exhibited a more protective	2012 [112]
	2Mo-(0.5-1.0)N	0.1 g/L of		°C	voltammetry	passive layer of higher thickness with	
	(HNS) and 317L	albumin and			(CV), Mott-	increase in nitrogen content. Passive	
		PBS with			Schottky	films showed n-type semiconductor	
		0.1g/L of				behavior and decrease in donor density	
		fibrinogen				was observed with increase in nitrogen.	
8.	N1 (18.4 Cr, 15.96	0.5 M NaCl +	-	-	EIS	It is suggested that the addition of	2015 [113]
	Mn, 2.23 Mo, 0.52 N,	0.5 M H <sub>2</sub> SO <sub>4</sub>				nitrogen accelerates the dissolution	
	0.048 C, 0.24 Si,					process, which is attributed to the	
	0.004 S, 0.017 P), N2					accumulation of passivation species and	
	(18.4 Cr, 15.96 Mn,					improvement in passivity.	
	2.23 Mo, 0.66 N,						
	0.048 C, 0.24 Si,						
	0.004 S, 0.017 P), N3						
1				1			

	(18.9 Cr, 18.9 Mn,					
	2.28 Mo, 0.96 N,					
	0.058 C, 0.2 Si, 0.003					
	S, < 0.03 P)					
9.	HNS (0.18-0.186 C,	Ringer's	6.5-7.5	Potentiodynamic	HNS showed higher pitting resistance	2018 [114]
	12.6-12.61 Mn, 17.4-	solution and		poalarisation, Pin	as compared to 316L under static as	
	17.5 Cr, 3-3.2 Mo,	Artificial		on disc	well as under applied load.	
	0.5-0.51 N, 0.06-0.07	saliva solution		tribocorrosion at		
	S, 0.013-0.014 P,			normal loads of 2		
	0.41-0.42 Si) and			and 10 N		
	316L					

Note: RT: Room temperature; PBS: Phosphate buffer saline; FCS: Fetal calf serum; MEM: Minimum essential medium; EIS: Electrochemical impedance spectroscopy; XPS: X-ray photoelectron spectroscopy; CPT: Crevice pitting temperature; MARC: Measure of alloying for resistance to corrosion

Appendix D: Effect of treatment of surface through surface mechanical attrition and ultrasonic shot peening on various properties of metallic materials.

S. No.	Alloys			Experimen	ntal details	Findings	References
		Shot	Peening	Frequency	Methods		
		size	Duration	(Hz)			
			(min)				
1.	AISI 316L	3	30	20k	Tensile tests of the	It exhibited an extremely high	2005 [144]
					nanocrystalline 316L stainless	tensile strength of 1450 MPa	
					steel at RT		
2.	AISI 316L	8	60	50	Corrosion in 0.05M H <sub>2</sub> SO <sub>4</sub> +	Improvement in corrosion	2006 [213]
					0.25M Na <sub>2</sub> SO <sub>4</sub>	resistance	
3.	AISI 316L	2, 3	15	20k	High cycle fatigue at R= -1 at	Increase in fatigue life and	2006 [158]
					10Hz frequency in air	fatigue limit with increase in	
						shot size	
4.	AIS 316L	3	30	20k	Ball on disc sliding wear test in	Increase in wear resistance	2006 [214]
					ambient condition	following USP	
5.	SS 400	8	15	50	High cycle fatigue at R=0 at	Fatigue strength increased by	2009 [215]
					40Hz frequency in air	13.1%	

6.	AISI 409	2, 5,	15, 30, 45	50	Corrosion in 0.6M NaCl	Increase in corrosion resistance	2010 [193]
		8				with 2 mm shots whereas	
						decrease with bigger shots	
7.	AISI 304	2, 5,	15, 30, 45,	50	Corrosion in 0.6M NaCl	No significant effect	2013 [194]
		8	60				
8.	IN 718	3	45, 60, 90	20k	Tensile test at RT	Marginal increase in strength	2014 [216]
9.	AISI 304	6	180	40	Tribocorrosion in 0.9% NaCl	Wear rate decreased by 2 to 4	2014 [217]
					solution	times following SMAT	
10.	AISI 304	2, 5,	15, 30, 45,	50	Corrosion in Ringer's solution	Increase in corrosion resistance	2015 [170]
		8	60			with 2 mm shots and decrease	
						with bigger shots	
11.	Al 2014	3	10	20k	LCF tests at total strain	A significant increase in fatigue	2015 [218]
					amplitudes of ±0.375 %, ±0.40	life.	
					%, ±0.50 % and ±0.55%. High		
					cycle fatigue at ±180 MPa.		
12.	Mg-6Gd-	8	3	20k	Dry sliding wear	Decrease in wear resistance	2015 [154]
	3Y-0.5Zr					after SMAT	
13.	Titanium	3	30	50k	Ability to new bone formation	There was a significant increase	2015 [131]
					is assessed by in vitro cell	in Osseo-integration following	
					adhesion, proliferation,	USP	

					differentiation and mineralisation		
14.	AISI 316L	5	2	20k	Cell culture, cell adhesion and proliferation, using human osteoblast cell (Saos-2)	Enhancement in cell attachment, spreading and proliferation rate following USP	2017 [129]
15.	AISI 316, AISI 690	3	15, 30, 45, 60	20k	Corrosion at RT and 300 °C in simulated steam generators (SG) environment.	Corrosion resistance decreased at RT, whereas it increased in simulated SG environment.	2017 [127]
16.	AISI 316L	3	_	20k	LCF tests at strain amplitudes of ±0.50 %, ±0.80 %, ±1.0 % and ±1.25 %	Decrease in fatigue life at a higher strain amplitude of $\pm 1.25$ whereas no effect at lower strain amplitudes.	2017 [165]
17.	Ti-6Al-4V	3	5	20k	LCF tests at strain amplitudes of ±0.60 %, ±0.65 %, ±0.70 %, ±0.80 %, ±0.90 %, ±1.0 %	Enhancement of fatigue life at lower strain amplitudes whereas no effect at higher strain amplitudes of $\pm 0.90$ % and $\pm 1.0$ %	2017 [132]
18.	Al 7075	3	0.25, 0.5, 1 and 5	20k	Corrosion resistance in 3.5% NaCl	Enhancement of corrosion resistance for 0.25 min of USP	2017 [133]

19.	Al 7075	3	0.5, 1, 3, 5	20k	LCF tests at strain amplitudes of	Increase in fatigue life with	2017 [134]
					±0.38 %, ±0.40 %, ±0.45 %,	increase in USP duration	
					$\pm 0.50$ %, $\pm 0.55$ %, $\pm 0.60$ %		
20.	MG-AZ31B	2, 3	15	20k	Corrosion fatigue in 3% NaCl	Improvement in corrosion	2019 [153]
						fatigue life following SMAT	
21.	Co28Cr6Mo	3	15	20k	Corrosion in Ringer's solution	Enhancement in corrosion	2018 [128]
						resistance	

Note: RT: Room temperature; R: stress ratio; LCF: Low cycle fatigue; HCF: High cycle fatigue; SMAT: Surface mechanical attrition treatment; USP: Ultrasonic shot peening

**Appendix E**: Effect of cold rolling and severe plastic deformation on various properties of the nickel free (or negligible nickel) nitrogen stabilised austenitic stainless steel.

S. No.	Alloys/Composition	Experimental details	Findings	References
1.	C-0.04, Cr-18.4, Mn-	Cold reduction in thickness was done to	A decrease in pitting potential was observed	2008, 2009
	15.8, Mo-2.19, N-0.66,	8, 30, 40, 49 and 60%. Potentiodynamic	in a solution of 3.5 % NaCl due to an	[219,220]
	Si-0.24, S-0.005, P-	and EIS tests were performed in 3.5%	imperfect and less protective passive layer	
	0.017	NaCl, 0.5 M NaCl + 0.5 M $\mathrm{H}_2\mathrm{SO}_4$ and	formed, attributed to the defects induced by	
		0.5 M NaCl + 0.5 M NaOH solution.	the cold rolling. However, no appreciable	
			reduction in corrosion resistance was	
			observed in 0.5 M NaCl + 0.5 M $H_2SO_4$ and	
			0.5 M NaCl + 0.5 M NaOH solution.	
2.	316L, 316LVM,	Samples were cold rolled for 10 and 20	There was no phase change, but strength and	2013
	316MnN <sub>1</sub> (0.34),	% thickness reduction. Tensile tests,	hardness increased with increase in cold	[105,206,221]
	316LMnN <sub>2</sub> $(0.43),$	hardness and microstructural study was	reduction. Nitrogen-containing steels	
	316LMnN <sub>2</sub> (0.52)	conducted. Cell adhesion and MTT assay	showed higher cell proliferation as	
		tests were also performed for DL cells to	compared to 316L and 316LVM. Cell	
		evaluate cytotoxicity. The corrosion	proliferation increased with cold reduction.	
		resistance in SBF environment was	Corrosion resistance increased with	
		evaluated at 37 °C using EIS and	increasing nitrogen content and cold	
		polarisation techniques.	working.	

3.	C <0.08, Si- 0.2-0.6,	Samples were cold compressed for 10,	Strength increased with increase in the cold	2014 [208]
	Mn-18-18.7, Cr-18-20,	20, 30, 40 and 50% at strain rate of	compression.	
	V< 0.2, N-0.51	0.05/sec at RT.		
4.	C-0.076, Mn-19.78, Cr-	USP was carried out with 3 mm shots at	There was decrease in fatigue life following	2014 [167]
	17.96, Si-0.34, N-0.543,	80 µm amplitude with 20kHz frequency	USP as compared to non-USP.	
	P-0.051, S-0.007, Ni-	for different duration. Low cycle fatigue		
	0.16, Mo-0.11, Cu-0.01	behaviour was studied following 10		
		minutes of USP.		
5.	00Cr18Mn15Mo2N0.86	Samples were cold deformed at different	There was no significant change in pitting	2016 [166]
		levels of 10, 20, 30, 40 and 50% and	potential.	
		electrochemical corrosion tests of		
		polarisation and EIS were performed in		
		0.9% saline at 37 °C.		
6.	C-0.014, N-0.9, Cr-18.3,	20% and 35% pre-strained fatigue	Fatigue and corrosion fatigue limit of non-	2018 [168]
	Mn-15.3, Mo-2.4, Si-	samples were tested in air and PBS	strained sample was found 550 MPa and 475	
	0.31, S-0.009, P-0.01	solution kept at 37 °C, at R=0.1 for $10^7$	MPa, respectively, and there was increase in	
		cycles.	fatigue limit and fatigue life with increasing	
			pre-straining.	

Note: RT: Room temperature; PBS: Phosphate buffer saline; EIS: Electrochemical impedance spectroscopy; USP: ultrasonic shot peening; R: Stress ratio

Appendix F: Glossary of words [222, 223, 224].

Acute: Duration of less than 30 days.

**Biocompatibility:** The ability of a material to perform with an appropriate host response in a specific application. It may also be defined as "biological performance of a material in a specific application that is judged suitable to that situation".

Biological environment: Conditions encountered within an animal or human body.

**Breakdown potential:** The potential at which there is breakdown of passive film and drastic increase in corrosion current with little increase in potential.

Capsule: Tissue surrounding an implant produced by local host response.

Cell culture: Growth of cells dissociated from the parent tissue by spontaneous migration or mechanical or enzymatic dispersal.

**Corrosion potential:** Potential at which rate of anodic dissolution of the electrode equals the rate of cathodic reaction. It is also called zero current potential or open circuit potential.

Cyclic stress amplitude: It is half of the algebraic difference between the maximum and minimum stress in a cycle.

**Cytotoxic**: Having a deleterious effect on cells.

**Cytotoxicity:** Cellular damage to one or more metabolic pathways, intracellular processes, or structures resulting in impaired function. Often, but not necessarily, linked to loss of viability.

Elastic strain amplitude, cyclic: It is half of the algebraic difference between the maximum and minimum elastic component of strain in a cycle.

**Endurance limit:** It is the stress limit below which materials are supposed to run for infinite number of cycles. In general, it is defined for  $10^7$  or  $10^8$  cycles of cyclic loading.

**Fatigue failure:** Materials subjected to a repetitive or fluctuating stress will fail at a stress much lower than that required to cause fracture on a single application of load. Failure occurring under such conditions of dynamic loading are called fatigue failure.

Frequency: Number of cycles per second

Hemocompatibility: Compatibility of material with blood.

Hemolysis: Release of hemoglobin due to damage to red blood cells.

**Implant:** A medical device made from one or more biomaterials that is intentionally placed within an animal or human body by the act of implantation.

**Implantation:** Placement of a device or material within the body of an animal of human by a medical or surgical professional, in such a way as to breach one or more epithelial layers and to leave materials and/or components in place after the initial procedure is completed.

In vitro: Literally, "in glass", but used conventionally to mean cultured outside of the host as cell cultures, organ cultures, or short-term organ bath preparations.

In vivo: In the living plant or animal.

Local Host response: The response, other than the intended therapeutic response, of tissue and organs contacting a biomaterial.

Passive film: The ultra-thin oxide layer formed on the surface which protect against corrosion.

**Physiological environment:** Controlled chemical (inorganic) and thermal conditions simulating a portion of a biological, biophysiological, or pericellular environment.

**Pitting:** Formation of spots or pits on the surface due to localised corrosion.

Plastic strain amplitude, cyclic: It is half of the algebraic difference between the maximum and minimum plastic component of strain in a cycle.

**Prosthesis:** A device that replaces a limb, organ, or tissue of the body.

Stress ratio (R): It is the ratio of minimum stress to maximum stress in a cycle.

Total strain amplitude, cyclic: It is half of the algebraic difference between the maximum and minimum strain in a cycle.

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#### List of Publications/Patents

### Patent

**Title:** A composition of austenitic stainless steel for medical implants, Indian Patent. Patent number 202011015355, Application date: 07.04.2020.

#### **International Peer-Reviewed Journals**

- Chandra Shekhar Kumar, Gaurav Singh, Suruchi Poddar, Neelima Varshney, Sanjeev Kumar Mahto, Arijit Saha Podder, Kausik Chattopadhyay, Amit Rastogi, Vakil Singh and Girija Shankar Mahobia, "High-manganese and nitrogen stabilized austenitic stainless steel (Fe-18Cr-22Mn-0.65N): a material with a bright future for orthopedic implant devices", Biomedical Materials, 2021, Vol. 16, PP. 065011.
- 2. Chandra Shekhar Kumar, Kausik Chattopadhyay, Vakil Singh and Girija Shankar Mahobia, "Enhancement in low-cycle fatigue life of high-nitrogen austenitic stainless steel at low strain amplitude through ultrasonic shot peening", Materials Today Communications, 2020, Vol. 25, PP. 101576.
- 3. Chandra Shekhar Kumar, Girija Shankar Mahobia, Arijit Podder, Sanjeev Kumar, Rahul Kumar Agrawal, Kausik Chattopadhyay and Vakil Singh, "Role of ultrasonic shot peening on microstructure, hardness, and corrosion resistance of nitrogen stabilised stainless steel without nickel", Materials Research Express, 2019, Vol. 6, Issue 9, PP. 096578.

#### **Conference Proceedings**

- 4. Chandra Shekhar Kumar, Girija Shankar Mahobia and Kausik Chattopadhyay, "Effect of ultrasonic shot peening on strain controlled fatigue behaviour of low nickel austenitic stainless steel", Materials Today: Proceedings, 2020, Vol. 28, PP. 808-812.
- 5. Girija Shankar Mahobia, Chandra Shekhar Kumar and Kausik Chattopadhyay, "Nanocrystallisation of nickel free high nitrogen austenitic stainless steel through ultrasonic shot peening", Key Engineering Materials, 2019, Vol. 813, PP. 43-48