# Chapter 2

# **Literature Review**

The effect of incorporation of various antibacterial agents as secondary phases, tailored structures (micro to nano-sized) and techniques such as, application of external stimuli (electrical and magnetic fields) for improving the antibacterial response has been reviewed. In addition, the effect of surface polarization and external electrical stimuli on cytocompatibility of various piezoelectric bioceramics has also been discussed.

#### **2.1 Introduction**

The development of suitable prosthetic implants for human health care is one of the stimulating areas in orthopedics [1]. Despite of significant progress in orthopedics, the research is in continuous thrust to address the number of existing issues which lead to the failure of prosthetic implants [2]. The bacterial infections at implant site have been recognized as one of the serious concerns [3]. Although, both, gram positive and gram negative bacteria are responsible for the microbial infection, maximum infection (~ 65%) is caused due to gram-positive cocci [Staphylococcus aureus (S. aureus) and Staphylococcus epidermidis (S. epidermidis)] [4]. However, gram negative bacteria [Escherichia coli (E. coli) and Pseudomonas aeruginosa (P. aerugenosa)] cause about 11 % of the bacterial infections in orthopedic implants [5]. Commonly used orthopedic implant materials such as stainless steel, cobalt-chromium, hydroxyapatite, bioglasses, tricalcium phosphate, and polyamide polymethyl methacrylate etc., are susceptible towards microbial infections [6,7]. To overcome the problem of implant associated infections, various techniques such as, modifications in surface chemistry, the compositional variation of prosthetic implants as well as the application of external stimulants are in continuous thrust. Recently, various piezobioceramics have been developed as prospective materials for orthopedic applications due to piezoelectric nature of bone [8,9]. The piezoelectric  $Na_{0.5}K_{0.5}NbO_3$  (NKN) and  $BaTiO_3$  demonstrate antibacterial response [10,11]. On the other hand, piezoelectric ceramics enhance the bioactive performance as well as cellular response [12]. The addition of piezoelectric material as a secondary phase in HA matrix enhances the proliferation of human osteoblast cells [13]. This chapter reviews various biomaterials and techniques for improving the antibacterial response. In addition, the cellular response of tailored biomaterials has also been reviewed.

#### 2.2. Antimicrobial response of tailored biocomposites

In this section, the effect of addition of various antibacterial agents as secondary phases on antibacterial response of composite systems has been discussed along with their mechanisms.

# 2.2.1. Effect of addition of Ag on antibacterial response of biocomposites

Ag possesses a good antibacterial property, which is used as an antibacterial agent since ancient time. [14,15]. It has been reported that the growth of *E. coli* bacteria is inhibited by ~ 62 %, 88 %, and 100 %, with addition of 10, 50, and 100  $\mu$ g/cm<sup>3</sup> Ag in HA, respectively, after incubation of 24 h [16]. In another study, it has been observed that the addition of 1000 ppm of Ag as secondary phase in HA increases the diameter of inhibition zone by about 41 and 46 % for *S. aureus* and *E. coli* bacteria, respectively. Pandey et al. [17] reported that the viability of *E. coli* and *S. aureus* bacteria reduces by ~ 61 % and 53 %, respectively, with the addition of 2.5 wt. % Ag in HA matrix.

The formation of reactive oxygen species (ROS) as well as  $Ag^+$  ions are responsible for the antibacterial response of Ag based composites [18]. The  $Ag^+$  ions are produced by its oxidation, these ions are highly reactive with bacteria [19]. The produced  $Ag^+$  ions can bind to DNA, RNA and proteins, present in bacterial cells to inhibit their growth [33]. The  $Ag^+$  ions distorted the structure of bacterial cells. On the other hand, the ROS such as, superoxide's, hydroxyl radicals, hydrogen peroxide radicals etc., interact with the cell membranes of bacteria and damage them [20]. Fig. 2.1 demonstrates consequences of the interaction of silver ions with the cell wall [21].



Fig. 2.1. Schematic illustrating the mechanism of antibacterial response due to silver ions. [Reproduced with permission from ref. (21): Copyright (2018), American Chemical Society].

# 2.2.2. Effect of addition of TiO<sub>2</sub> on antibacterial response of biocomposites

TiO<sub>2</sub> is a non-toxic metal oxide with good antibacterial property [22]. Recently, TiO<sub>2</sub> has been used as secondary phase in base materials (HA) for improving the antibacterial response of develpoed composite system [32]. The diameter of inhibition zone for HA-60 wt. % TiO<sub>2</sub> has been increased by ~ 31 and 36 % as compared to control sample, while cultured with *S. aureus* and *E. coli* bacteria, respectively [23]. It has been reported that colony formation unit (CFU/ml) of *bacillus subtilis* bacteria decreases by ~ 99.90 % on AgCl/TiO<sub>2</sub> composite, while exposed to visible light for 3 h. However, for *pseudomonas aeruginosa* bacteria biofilm formation has been reduces to 57 % for the same substrate and similar exposure conditions [24].

It has been demonstrated that  $TiO_2$  oxidises in presence of visible light and produces free radicals such as hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>), superoxide's, peroxides (O<sub>2</sub><sup>-2</sup>), hydroxyl

radicals (OH<sup>-</sup>) etc. These free radicals are responsible for antibacterial response of TiO<sub>2</sub> based bio-composite systems [25,26]. TiO<sub>2</sub> shows photo-catalytic behavior under UV light, however, surface modification of TiO<sub>2</sub> (with iodine) enhances the visible absorbance [27]. In presence of visible light, TiO<sub>2</sub> surface, irradiated by photon energy, produces electron (e<sup>-</sup>) and hole (h<sup>+</sup>) pairs at conduction and valence bands, respectively [Eq.(2.1)]. The generated electrons and holes have strong oxidizing power. These electrons and holes may recombine and produce ROS as shown in equations (2.2) and (2.3), these radicals can damage the outer layer of bacteria. [28]. Fig.2.2 schematically represents the mechanisms, responsible for the antibacterial activity of TiO<sub>2</sub> based biocomposites [29].

$$\Gamma iO_2 + h\nu \rightarrow e^- + h^+ \qquad (2.1)$$

$$\mathbf{0}_2 + \mathbf{e}^- \to \mathbf{0}_2^- \tag{2.2}$$

$$H_2O + h^+ \rightarrow (OH)^- + h^+$$
 (2.3)



Fig. 2.2. Schematic diagram of proposed mechanisms for antibacterial response of TiO<sub>2</sub> [Reproduced with permission from ref (29): Copyright (2014), Royal Society of Chemistry].

#### 2.2.3. Effect of addition of ZnO on antibacterial response of biocomposites

In general, few of the metal oxides such as, ZnO, MgO, CuO etc., illustrate antibacterial effect [30]. In recent years, number of biocomposites with ZnO is recognized for their potential biocompatibility, biodegradability and antibacterial nature [31, 32]. The colonies of *E. coli* bacteria reduce by ~ 13 % to 50.45 % for HA- ZnO biocomposite while increasing the ZnO content by 1.5 % to 30 wt. %, after incubation of 4 h. However, the incubation of 24 h reduces the colonies of *E. coli* bacteria by ~ 26 % to 60 % for similar material composition and bacteria [33]. The incorporation of 25 wt. % ZnO in HA reduces the biofilm formation by ~ 98 % and 99 % while cultured with *S. aureus* and *E. coli* bacteria, respectively [34].

The release  $Zn^{2+}$  ions and generation of ROS are responsible for antibacterial response of ZnO based bio-composites [35]. The ROS and  $Zn^{2+}$  ions are toxic in nature which can destroy the outer membranes of bacterial cells [36]. The ROS reacts with lipid layer of bacterial cell (for gram-negative bacteria) and destroy cell wall structure and consequently, the death of bacterial cell [37]. On the other hand,  $Zn^{2+}$  ions diffuse in the cell walls (for gram-positive bacteria) and disrupt the amino acid metabolism and enzymes which leads to cell death [38-40]. Fig. 2.3 illustrates the mechanisms, responsible for antibacterial response of ZnO containing composites [41].

The effect of incorporation of various antibacterial agents on antibacterial response of HA has been summarized in Table 2.1.



Fig. 2.3. Schematic presentation of mechanisms for antibacterial response of ZnO based biocomposites. [Reproduced from ref. (41): open access]

Table 2.1 Effect of addition	of secondary phase on	antibacterial response of HA
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	Material Composition		Pathogen	Effects	Ref.	
S.N.	Base material	Secondary phase	(Incubation Time :12 h)			
1.	НА	Ag	S. mutans S. anguinis	Diameter of inhibition zone 6.33 mm (for 5wt.% Ag in HA) 8.66 mm (for 10 wt.% Ag in HA) 7.66 mm (for 5wt.% Ag in HA) 9.66 mm ( for 10 wt.% Ag in HA)	6	
			L. acidophilus	5.66 mm (for 5wt.% Ag in HA) 7.66 mm ( for 10 wt.% Ag in HA)		
2 НА 1000 рг		1000 ppm of	S. aureus	17 mm		
2.		Ag	Pneumococcus	17.5 11111	16	
			E. coli	18 mm		

3.	НА	TiO <sub>2</sub> (10-60 wt.	E. coli	14 mm (60 wt.% TiO <sub>2</sub> )	33
		%)	S. aureus	21 mm (60 wt.% TiO <sub>2</sub> )	
4.	НА	ZnO (1.5-30 wt. %)	E. coli	Bacterial colonies decreased by 35% (for 10 wt.% ZnO) 60 % (for 30 wt. % ZnO)	
_		-	E. coli	14.7 mm	10
5.	НА		S. epidermidis	10 mm	42
6.	НА	Collagen/pol y lactic acid with antibiotic vancomycin (VCM/nHA C/PLA)	<i>S. aureus</i> (72 h)	21.07 mm	43
	Fluro Hydroxy apatite (FHA)	FHA -40Zr- 20Ce composite. (Compositio ns are in wt. %)	S. aureus	42 mm	
			E. coli	10-13 mm	
	ZnO	Cr	S. aureus	15-17 mm	
7.	ZnO		S. aureus	6-9 mm	44
		_	E. coli	No inhibition zone found	
8.	НА	AgCl/ TiO <sub>2</sub>	<i>E. coli</i> (for 3 h)	Colony forming unit (CFU) decreases by 99.10 % when exposed to visible light irradiation	45
9.	FHA	Zr- Ce (40	E. coli	Diameter of inhibition zone	46

		wt.% Zr &		37 mm	
		20 % (CC)	S. aureus	42 mm	
		0.2 wt. % Zn,	E. coli	15 mm	
10.	HA	Ag, and	S. aureus	16 mm	47
		0.025 wt. % Au	Bacillus cereus	14 mm	

#### 2.3. Effect of nanostructured materials on antibacterial response

Nanomaterials gained a good attention in biomedical field due to their higher surface area, catalytic activity, dissolution rate etc. [48]. They demonstrate different properties than their respective counter bulk materials. The nano structured Fe, TiO<sub>2</sub>, and ZnO etc. possess good antibacterial properties as compared to their bulk counterpart [49]. In another study, it has been found that  $Fe_2O_3$  nanoparticles show antibacterial activity whereas such response in its bulk counterpart is not observed [50].

#### 2.3.1. Antibacterial response of zinc oxide (ZnO) nanoparticles

The nanostructured ZnO have been used in biomedical fields for bio-sensing, imaging, drug delivery and bio-implant applications [51]. The nano structured Zn demonstrates excellent antibacterial properties [52]. The antibacterial activity of ZnO nanoparticles increases with reducing the particle size, even in nano-size range [53]. The content of ZnO in the solution/base materials is the influential factor for improving the antibacterial property [54]. The colonies of *E. coli* bacteria on ZnO nanoparticles (~ 40 nm) reduced by ~ 86 % after incubation of 24 h [54]. However, for *K. pneumonia*, the bacterial colonies have been decreased by ~ 75.38 % at the concentration of 0.75 mM for similar material and incubation period [47]. The antibacterial activity of Ti-ZnO nanorods increased by 65.5 % and 55.4 %, against *S. aureus* and *E. coli* bacteria, respectively, after incubation of 12 h [55]. However, the hybrid ZnO/polydopamine (PDA) / arginine-

glycine-aspartic acid crystine (RGDC) nanorods have found to be ~ 72.2 % and 74.7 % efficient against *S.aureus* and *E. coli* bacteria, respectively, while incubated with similar condition [56].

The release of zinc ions, owing to the dissolution of ZnO as well as formation of ROS are the suggested mechanisms responsible for the antibacterial activity of ZnO nanoparticles [57]. The nanosized ZnO particulates penetrate the cell wall easily and produces comparatively higher amount of ROS than that of bulk-sized ZnO [58]. The point defects in nanoparticles increase the abrasive nature of its surface which injured the bacterial cells [59]. However, surface defects, produced by the partial dissolution of nanosized ZnO in water, result in an uneven surface texture that damage the bacterial cells [60]. In another study, it has been demonstrated that physical interaction of ZnO nanoparticles with the bacterial cells also play an important role in bactericidal effects [61]. The strong electrostatic interaction takes place between bacterial cell surface (negative charge) and  $Zn^{2+}$  ions [62]. As a result, ZnO nanoparticles ( $\leq 10$  nm) accumulate at the outer layer of plasma membranes and neutralize the surface charge of bacterial cells [56]. This leads to increase in surface tension, membrane permeability, change in membrane texture, morphology, and generation of oxidative stress which results in bacterial cell death [63]. The ZnO nanoparticles ( $\leq 10$  nm) passage through the cytoplasm membrane and accumulate inside bacterial cells (via particle internalization) which damage the intercellular constituents, including nucleic acids etc. [64,65]. The possible mechanisms for antibacterial response of ZnO nanoparticles are illustrated in Fig. 2.4 [62].



Fig. 2.4. Schematic illustration represents (a) various modes of antibacterial response due to ZnO nanoparticles and (b) possible mechanisms for ZnO nanoparticles mediated antibacterial response.[Reproduced from ref (63): open access].

# 2.3.2. Antibacterial response of silver (Ag) nanoparticles

Ag is used as an antibacterial agent since ancient time due to its inherent antibiotic properties. [66]. The nano-structured Ag has better antibacterial properties as compared to their bulk counterpart. [67]. Recently, Ag nanoparticles have been introduced as a quite appealing choice for the development of new generation bactericidal material [68]. The antibacterial response of Ag nanoparticles depends upon particle size as well as shape [69,. In addition the concentration of Ag in growth medium and exposure time are also important factors for improving the antibacterial response [70]. In another study, it has been reported that triangular shape Ag nanoparticles has better antibacterial response (as compared to other shapes) even at low concentrations [71]. The growth of *E. coli* bacteria in Luria-Bertina growth media has been reduced by about 70 % and 100 %, respectively, while increasing the concentration of Ag nanoparticle (~12 nm) from 10  $\mu$ g/cm<sup>3</sup> to 60  $\mu$ g/cm<sup>3</sup>, after incubation of 24 h at 37 °C [72]. However, 100  $\mu$ g/ml concentration of Ag nanoparticle inhibit the growth of *S. aureus* bacteria completely [73]. The viability of *E. coli* and *S. aureus* bacteria reduced to 96.3 % and 99.4 %

respectively, while cultured on graphene oxide and silver nanoparticles composite [74]. The antibacterial response of nanoparticles against various pathogens is summarized in Table 2.2.

The mechanism, responsible for the antibacterial response of Ag nanoparticles is very complicated. Catalina et al. [75] proposed three mechanisms for the antibacterial response of Ag nanoparticles. (i) The release of silver ions that disrupts ATP production and DNA replication. (ii) Formation of reactive oxygen species (ROS) and (iii) Direct damage of cell membrane by silver nanoparticles due to its inherent antibacterial property.

Ag nanoparticles interact with bacterial cells and penetrate inside (particle internalization) through pits and holes which disrupt the cell walls [Fig. 2.5 (a and b)]. It has been hypothesized that  $Ag^+$  ions are released by dissolving silver nanoparticles in growth media (Eqs. 2.4 and 2.5). [76].

$$2Ag + H_2O_2 + 2H^+ \rightarrow 2Ag^+ + 2H_2O, E^0 = 0.17V \qquad (2.4)$$

$$4Ag + O_2 + H_2O \to 4Ag^+ + 4OH^-$$
 (2.5)

Choi et al. [77] reported that silver nanoparticles oxidize in presence of oxygen and release silver ions as well as produce hydroxyl radicals. These silver ions interact with peptidoglycan layer in the bacterial membrane and consequently, disrupt the cell metabolic processes. This mechanism is dominated in case of gram-positive bacteria, owing to their thick peptidoglycan layers as compared to gram-negative bacteria [78]. In addition, the formation of ROS also leads to the death of bacterial cells [79]. Fig. 2.5 (c and d) represents the schematic diagrams of mechanisms, responsible for the generation of Ag ions [62].



Fig. 2.5. Schematic illustrating (a) nanoparticles internalization into the cell wall, (b) disruption of cell wall and release of intracellular materials, (c) thickenings of cell wall and release of cytoplasm, and (d) mechanisms of dissolution of nanoparticles in the bacterial membrane, release of metal ions, ROS, and disruption of DNA. [Reproduced from ref (63): open access]

# 2.3.3. Antibacterial response of copper and copper oxide nanoparticles

Recently, nano-sized Cu and CuO attracted attention due to their better antibacterial response as compared to its bulk counterpart [80]. However, the application of Cu nanoparticles is limited due to its rapid oxidation during air exposure [81]. The antibacterial response of Cu and CuO nanoparticles depends upon on the concentration of Cu in growth media for both, gram-positive and gram-negative bacteria [82]. The diameter of inhibition zone for *E. coli* and *S. aureus* bacteria has been increased by ~ 43 % and 48 % while increasing the concentration of Cu nanoparticles (~20 nm) by 25 % respectively, after incubation of 24 h [83]. In another study, it has been demonstrated that viability of *E. coli* bacteria reduced by ~ 48 %, after incubation of 12 h, while cultured on Cu nanoparticles (~ 20 nm) [84]. Carboxyl methyl chitosan (CMC) alginate

(Alg) scaffolds with Cu nanoparticles (CMC/Alg/Cu) shows better antibacterial response against *S. aureus* bacteria as compared to carboxyl methyl chitosan alginate (CMC/Alg) after incubation of 2 and 4 days [85]. Table 2.2 summarizes the antibacterial response of CuO nanoparticles against various pathogens.

It is assumed that Cu ions, produced by Cu nanoparticles, interact with sulfydryl groups that destroy the bacterial cells, enzymes and proteins [86]. The interaction of bacterial cells with Cu nanoparticles affects the membrane integrity due to decrease in transmembrane potential [87]. The nano-structured CuO produces more Cu<sup>+</sup> ions and ROS as compared to its bulk form [88]. It has been reported that CuO nanoparticles are more toxic than bulk CuO due to the solubility of CuO nanoparticles and release of Cu ions [89]. The released Cu ions interact with the bacterial cell membrane and generate oxidative stress, which damage the outer cell membranes and proteins of bacteria [90]. ROS production starts with the reduction of O<sub>2</sub> and synthesis of sulphoxide anion. ROS interact with the cellular membranes that lead to the damage of DNA, enzymes and other proteins [91]. Fig. 2.6 demonstrates the mechanisms responsible for antibacterial response of Cu nanoparticles [87].



*Fig.* 2.6. *Schematic illustration of antibacterial response due to Cu nanoparticles.*[*Reproduced from ref* (87): *open access*]

# 2.3.4. Antibacterial response of iron oxide nanoparticles

Iron oxide has been widely used biomedical field due to its reasonable antibacterial properties [92]. The antibacterial response of iron oxide nanoparticles also depends upon the particle size, concentration of iron oxide in growth media as well as exposure time [93]. It has been demonstrated that the size of inhibition zone increases by 61.53 %, 60.17 % and 55.17 % for *E. coli*, *P. aerugenosa*, and *S. aureus* bacteria, respectively while increasing the concentration of iron oxide nanoparticles (~10 nm) from 0.01 to 0.15 mg/ml in growth media, after incubation of 24 h [94]. However, the size of inhibition zone has increased by 21.42 % and 29.41 % against *E. coli* and *S. aureus* bacteria, respectively when standard antibiotics neomycin (30  $\mu$ g/disc) was used with iron oxide nanoparticles (~ 66 nm and concentration 50 mg/ml) and incubated for 24 h [58]. The CFU of *S. aureus* bacteria for porous iron-carboxylate metal-organic framework [MOF-53 (Fe)] nanoparticles decreases by 16.26 % while concentration of [MOF-53(Fe)] Fe ions increases to 87.5 %, after incubation of 24 h However, CFU reduction for vancomycin loaded MOF-53 (Van) decreases by 99 % while concentration

of vancomycin increases to 87.7 % for similar bacteria and incubation conditions [95]. The antibacterial response of iron-oxide nanoparticles with various pathogens are summarized in Table 2.2.

The mechanisms responsible for the antibacterial response of iron oxide nanoparticles are similar to other metal oxides such as ZnO,  $TiO_2$  and CuO etc. [96]. However, the primary cause of antibacterial response of iron oxide is oxidative stresses, produced via the formation of ROS which includes superoxide's, hydroxyl radicals, hydrogen peroxide and singlet oxygen molecules. ROS can be generated by Fenton reaction as [97],

$$Fe^{2+} + H_2O_2 \rightarrow Fe^{3+} + HO^{-} + OH^{-}$$
 (2.6)

$$Fe^{3+} + H_2O_2 \rightarrow Fe^{2+} + HOO + H^+$$
 (2.7)

ROS is toxic in nature which damages the proteins and DNA in the bacteria [98].

#### 2.3.5. Antibacterial response of magnesium oxide (MgO) nanoparticles

Magnesium (Mg) is a quite appealing biodegradable implant [99,100]. The nanostructured MgO exhibit anti-oxidation and anti-inflammatory properties, due to which, it is potentially used as antibacterial agent [101]. In general, nano-sized inorganic metal oxides have potential to kill bacteria because they react with intracellular oxygen and produce ROS [102]. The antibacterial response of MgO nanoparticles depends upon the annealing temperature, concentration of MgO and incubation time [103]. It has been reported that viability of *E. coli* bacteria decreases by 83 % and 95 % with the exposure of MgO nanoparticles (~ 8 nm) for 1 and 4 h, respectively. Similarly, for *S. aureus* bacteria, it has been reduced by 82 % and 99.9 %, respectively, for the similar particles and exposure conditions [104]. The antibacterial efficiency of Mg alloy (AZ 31) doped with silver nanoparticles and polymethyl trimethoxysilane (PMTMS) against *S. aureus* bacteria increases by 85 %, after incubation of 16 h. However, it is increased by 98.40 % for silver nanoparticles dipped in Polyethylenimine (PEI) solution for similar bacteria and incubation condition [105]. Sundrarajan et al. [106] reported that the diameter of inhibition zone for nano-structured MgO samples decreases with increasing the annealing temperature. The inhibition zone for *S. aureus* and *E. coli* bacteria has been found to be 23 and 21 mm on MgO nanoparticles (~ 4.6 nm), annealed at 300 °C.

Despite of a number of proposed mechanisms, the exact mechanism responsible for the antibacterial response of MgO nanoparticles is still unclear [107]. It has been hypothesized that oxidation and reduction reactions occur at the surface of MgO nanoparticles which produce ROS [108]. Peter et al. [109] suggested that antibacterial response of MgO nanoparticles is due to the presence of defects (oxygen vacancies) on the surface of nanoparticles. MgO nanoparticles hydrated to form Mg (OH)<sub>2</sub> at the surface that leads to the generation of electrons and holes. The oxygen molecules present on the surface react with electrons and produce superoxide's which leads to the destruction of cells. Due to hygroscopic nature, MgO nanoparticles absorb moisture and form a thin water layer around it. The local pH of this thin layer is higher than its equilibrium value in the growth media. When nanoparticles come in contact with bacterial cells, the higher value of pH damage the membrane which results in cell death [110,111]. Therefore, MgO nanoparticles induced pH variation of the growth media is also one of the possible reasons for the antibacterial response of MgO. Fig. 2.7 represents the mechanisms for antibacterial response of MgO nanoparticles [112].



Fig. 2.7. Schematic diagram representing the (a) interaction of bacterial cells with MgO and generation of ROS and (b) protein disruption with pH variation. [Reproduced from ref (112): open access]

# 2.3.6. Antibacterial response of gold (Au) nanoparticles

Gold nanoparticles have widely used in various biomedical applications such as imaging, gene therapy, water remediation, drug delivery, as well as antibacterial agents [113]. The size of inhibition zone increases by 33.33 % and 29.31 % against *S. aureus* and *E. coli* bacteria respectively, for Au nanoparticles with the concentration of 50  $\mu$ g/ml in media after incubation of 24 h. However, size of inhibition zone decreases by 29.41 % against *Basilus subtilus* bacteria with the concentration of 75  $\mu$ g/ml for similar particles and exposure time [114]. Table 2.2. summarizes the antibacterial response of gold nanoparticles with various bacterial cells.

There are basically two mechanisms, which are responsible for the antibacterial activity of Au nanoparticles. One of them is by collapsing the membrane potential that inhibits ATPase activity and secondly by inhibiting the subunit of ribosome for tRNA binding [115]. In contrast to the earlier discussed antibacterial agents, Au nanoparticles showed ROS independent antibacterial response [116]. Fig. 2.8 demonstrates the mechanisms for antibacterial response of Au nanoparticles [117].



Fig. 2.8. Schematic illustrating the possible mechanism for antibacterial response of gold nanoparticles. [Reproduced with permission from ref (117): Copyright (2015) Springer Nature]

S.N.	Nonomaterials	Average particle size	Pathogens (Incubation time: 12 h)	Concentration in growth medium		Effects	Ref.
1.	ZnO nanoparticles	6.8 nm	E. coli	0.1 mg/mL	Bacte	rial cell death 99.8 %	74
			S. aureus	0.1 mg/mL		98 %	
				FeeO.	Di inh	iameter of ibition zone (mm)	
2	Iron oxide	66 nm		concentration	With out antib iotic	With standard antibiotic	107
۷.	nanoparticles	00 1111	E. coli	50 mg/mL	11	14	107
			S. Epidermidi s	50 mg/mL	14	15	
			S. aureus	50 mg/mL	12	16	
			Bacillus subtilis	50 mg/mL	20	16	
3.	Silver nanoparticles	12 nm	E. coli	10 $\mu$ g/cm <sup>3</sup> 60 $\mu$ g/cm <sup>3</sup>	bact in	terial growth hibited by 70 % 100 %	81
			S. aureus	$100 \mu g/cm^3$		100 %	
		48 nm	B. subtilis	10 ppm		90 %	
	ZnO	cles	E. coli	1000 ppm		48 %	
4.	nanoparticles		S.aureus	0.1 mg/mL		98 %	118
			E. coli	550 μg/mL		100 %	-
			S.aureus	60 µg/mL		100 %	-
5	Ag-ZnO	64 nm	E. coli	0.01-0.15 mg/mL		61 %	119
5.	nanocomposites	nanocomposites 04 IIII	P. aerugenosa	0.01-0.15 mg/mL		60 %	
			S.aureus	0.01-0.15 mg/mL		55.17 %	
			E. coli	10-50 mg/mL		30 %	
6.	Iron oxide nanoparticles	35 nm	S.aureus	10-50 mg/mL		25 %	120
7.	TiO <sub>2</sub> nanoparticles	25 nm	E. coli 35	20-100 μg/mL		41.17 %	121

			-			
8.	TiO <sub>2</sub> nanoparticles	22.41 nm	E. coli S. epidermidis	100 mg/mL 144 mg/mL	Diameter of inhibition zone (mm) 12 15	122
9.	CuO nanoparticles	7.8 nm 4.8 nm	E. coli S.aureus E. coli S.aureus	103 μg/mL 120 μg/mL 103 μg/mL 120 μg/mL	15.8 13.2 16.8 15	123
10.	MgO nanoparticles	43-91 nm	S.aureus P. aerugenosa	1 mg/mL 1 mg/mL	27 24	124
11.	MnFe <sub>2</sub> O <sub>4</sub> / Ag	216 nm	E. coli S.aureus	1 mg/mL 1 mg/mL	35.8 42.8	125
12	Silver nanoparticles	16 nm	E. coli	60 μg/mL	Completely inhibit the CFU	126
13	Silver nanoparticles	14 nm	B. subtilis	13.5 µg/mL	76 % reduction in CFU	127
14.	CuO nanoparticles	23.17 nm	E. coli S.aureus P. aerugenosa	31.25 μg/mL 125 μg/mL 62.5 μg/mL	Completely inhibit the bacterial growth Completely inhibit the bacterial growth Completely inhibit the bacterial growth	128
				0.01.0.15	Diameter of inhibition zone increased by	
15.	Iron oxide nanoparticles	Iron oxide nanoparticles 10 nm	E. coli P. aerugenosa	0.01-0.15 mg/mL 0.01-0.15 mg/mL	61 % 60 %	129
			S.aureus	0.01-0.15 mg/mL	55.17 %	

# 2.4. Effect of external stimuli on antibacterial response

Apart from and nanoparticles and incorporation of secondary phase in base materials, the application of external stimuli such as magnetic and electric fields are the potential alternatives for reducing the bacterial infections [130-132].

#### 2.4.1. Effect of magnetic field on antibacterial response

The application of magnetic field has been recognized in various biomedical fields, such as cancer therapy and drug delivery as well as antimicrobial agent [133]. The intensity of applied magnetic field as well as exposure time is the main influencing factor to enhance the antibacterial response [134]. It has been reported that low-intensity DC magnetic field (2.7-10 mT) is not bacteriostatic on gram-negative E. coli bacteria [101]. However, the growth of S. aureus bacteria decreases with the exposure to low-intensity DC magnetic field (0.1-0.3 mT) [135]. The CFU for E. coli bacteria decreases by ~ 57 %, 68 %, and 76 % while exposed to static magnetic fields of 30, 50 and 80 mT, respectively, for 24 h. However, the CFU of S. aureus bacteria decreased by ~ 18 %, 32.47 %, and 61.73 %, respectively, for similar treatment (magnetic field and exposure time). In contrast, the growth rate of *bacillus subtilis* bacteria increases by 72 %, 75 %, and 81 %, respectively, after exposure to similar magnetic field intensity and time duration [136]. It has been reported that inhomogeneous (5.2~6.1T and 3.2~6.7 T) magnetic field is more effective than homogeneous (7T) magnetic field, as far as the viability of E. coli cells in the magnetic field is concerned [137]. The viability of bacterial cells decreases by 60 % and 70 % for S. aureus and E. coli bacteria, respectively, after exposure to the pulsed magnetic field of 4T for 30 mS [138]. Almost similar effect has been reported for exposure to the static magnetic field on antibacterial response, while both, gram-positive and gram-negative bacteria were cultured on various bio-ceramics surfaces. The viability of both, gram positive and gram negative bacteria, adhered on the surface of HA and HA-Fe<sub>3</sub>O<sub>4</sub> composites, decreases with increasing the duration of exposure (0.5-4 h) in the magnetic field (100 mT) as well as the  $Fe_3O_4$  content in the bio-composites [139]. There are basically two mechanisms which are responsible for antibacterial response due to the application of the external magnetic field. Fig.2.10 schematically represents the

mechanism of antibacterial response in the magnetic field [139]. Electromagnetic field affects the permeability of the ionic channels in the cell membrane. The other possible effects include the formation of free radicals such as, hydroxyl radicals, superoxide's etc., due to magnetic field exposure that increases the production of reactive oxygen species (ROS) which is bactericidal [140.-..144.].



Fig. 2.9: Schematic representation of mechanism for antibacterial activity due to production of free radicals in magnetic field. [Reproduced with permission from ref (139): Copyright (2015) Springer Nature]

# 2.4.2. Effect of electrical stimuli on antibacterial response

The external electric field induced antibacterial response depends on the intensity of electric field, exposure time and pulse duration (in case of pulsed electric field) etc. [145, 146]. The number of CFU for *E. coli* and *S. aureus* bacteria decreases by 33 % and 31 %, respectively, after exposure to the static electric field of 4.5 kV/cm for the duration of 30 min. However, CFU decreases by 56 and 54%, respectively, for similar bacteria and electric field, after exposure for 2.5 h. The antibacterial response also depends upon the frequency of the applied electric field. Mirzaii et al. [147] reported that the number of

CFU decreases by 3 % and 47.17 % against *S. aureus* bacteria, upon exposure to an electric field with strength of 6 V/cm<sup>2</sup> and frequencies of 1 MHz and 20 MHz respectively, for 6 h. However, the similar treatment (electric field and frequency) leads to the reduction of CFU by 10 % and 27 % for *P. aerugenosa* bacteria, when exposed for 4 h. Table 2.3 summarizes the antibacterial activity for both, gram positive and gram negative bacterial cells under the exposure to the electric field. Pulsed electric field (PEF) is used for killing the bacteria in food preservation. Malicki et al. [148] reported that the maximum reduction (4.7 log unit) in CFU/ml against *E. coli* bacteria is obtained after the exposure of pulsed electric field (32.89 kV/cm) with 180 pulses for the duration of 30  $\mu$ s. However, CFU/ml decreasesby 1.2 log unit against *bacillus cereus* bacteria, through exposure of 16.7 kV/cm PEF for 2 mS (50 pulses) [149].

The mechanism, responsible for electric field induced antibacterial activity, includes electrolysis of molecules on the surface of bacterial cells that produce the toxic substances like  $H_2O_2$ , oxidizing radicals, and chlorine molecule [150,151]. Fig. 2.10 demonstrates the mechanisms, suggested for antimicrobial response in electric field [152].

The following reactions occur at cathode and anode [153]. Production of  $H_2O_2$  at cathode,

$$H_2O + 2e^- \rightarrow H_2 + 2OH^-$$
 (2.8)

$$H_2 + 2H^- + 2e^- \rightarrow H_2O_2$$
 (2.9)

Production of Cl<sub>2</sub> at anode,

$$\begin{array}{ll} 2H_2 0 \to 0_2 + 4H^+ + 2e^- & (2.10) \\ \\ 2Cl^- \to Cl_2 + 2e^- & (2.11) \end{array}$$

The application of electric field induces a potential across the cell membrane for short duration which causes the loss of membrane resistance [154,155]. The strength of the applied electric field is also an important parameter for pulse electric stimulation induced bactericidal effect. Usually, the electric field of strength of 20-80 kV/cm is applied for a short duration (generally in microseconds) for killing the bacterial cells [156]. The mechanism, responsible for antibacterial response due to the application of electric current is the generation of chemical oxidant as well as the hydrophilicity of suspension medium [157-.159]. Application of electric current to bacterial suspension generates few chemical oxidants such as chloride on electrodes due to electrolysis, which is bactericidal [160]. Oxidative stress develops in the solution due to ions which cause the significant reduction in hydrophilicity i.e., reduction in biofilm formation [161]. A high-intensity electric current can alter the orientation of membrane lipids and also oxidize the cellular constituents which destruct the cell membrane [162].



Fig. 2.10: Schematic representation for the mechanisms of antibacterial response due to electric field. [Reproduced with permission from ref. (21): Copyright (2018), American Chemical Society].

 Table 2.3: Effect of incorporation of secondary phase as well as external stimulant on

	Mate	erial	External			
S.N	Base material	Second ary phase	stimulant (Magnetic field/ electric field/ surface charge)	Types of pathogens and exposure time	Effects	Ref.
1.	НА	-	Static Magnetic field of 100 mT	<i>E. coli</i> for 4 h <i>S. epidermidis</i> for 4 h	Cell viability decrease by 60% 17 %	163
2.	НА	Fe <sub>3</sub> O <sub>4</sub> (10-40 wt. %)	Static Magnetic field of 100 mT	<i>S. aureus</i> for 4 h	Bacterial adhesion is reduced by 8% ( HA) 25% (HA-40 wt.% Fe <sub>3</sub> O <sub>4</sub> )	164
3.	Glass substrate	_	Low-intensity electric field (1.5-30V/cm)	S. aureus for 4h	Bacterial adherence Recesses to 98% No effect on	165
4.	НА	ZnO (10 wt. %)	DC electric field of intensity 1V/cm	<i>S. aureus</i> for 24 h	bacterial adhesion %Viability reduction 60 % (pure HA) 70% (10wt.% ZnO in HA)	166
5	НА	-	Positively charged surface	S. aureus for 72h	Bacterial cells per colony decreases by 42.10 %	167
			Negatively	S.aureus for 72h	28 %	

antibacterial response of bioceramic substrates

			charged surface			
				S. epidermidis 4 h	No effect on bacterial adhesion	
6.	Glass ceramic (Cerec Blocs S3- M14)	-	Polarized surface (DC electric field of 0.1 -1 kV/cm in air at 250 °C for 1 h)	<i>Streptococcus mutans</i> for 24 h	Reduction in bacterial adhesion.	168

# 2.5. Effect of addition of piezoelectric secondary on antibacterial response of bioceramics

It has been reported that piezoelectric materials demonstrate antibacterial response [169]. Recently, various piezobioceramics are emerged as prospective materials for bone implant, due to the piezoelectric nature of bone [170]. The piezoelectric  $Na_{0.5}K_{0.5}NbO_3$  (NKN) and BaTiO<sub>3</sub> demonstrate antibacterial response [171]. It has been reported that incorporation of BT (40 and 60 wt. %) in HA matrix increases the diameter of inhibition zones by (123,154 %) and (136,140 %) for *S. aureus* and *E. coli* bacteria, respectively [172]. The viability of *S. aureus*, *E. coli* and *P. aeruginosa* bacteria reduces by almost 43, 35 and 34 % on HA - 40 wt. % BaTiO<sub>3</sub>, respectively. However, the addition of 60 wt. % BaTiO<sub>3</sub> in HA reduces the viability by 56, 66 and 51 %, while cultured with similar bacteria and incubation condition [173]. Overall, the litrature reports suggested that the incorporation of piezoelectric secondary phases in HA increases the antibacterial properties.

#### 2.6. Cellular response of tailored biocomposites

Apart from antibacterial properties, the cytocompatibility of developed implant materials is very serious concern. In this section the cellular response of various bioceramics as wll as effect of external stimuli on cellular response have been reviewed. Saha et al. [174] reported that the addition of ZnO (< 7.5 wt. %) in HA matrix increases the cellular proliferation. The adhesion of FMM1 fibroblasts cells increases by 150 % on tri calcium phosphate (TCP) – 50 wt. % HA after incubation of 3 days [175]. Rajendran et al. [176] reported that viability of NIH3T3 cells on HA-10 wt % Ag, increases approximately 80 %, after incubation of 7 days. Wu et al. [177] reported that cytocompatibility of MG 63 cells on magnesium calcium phosphate cement (MCPC) was higher than that of calcium phosphate cement (CPC) and magnesium calcium phosphate cement (MPC) while incubated for 7 days. The optical density of MG-63 cells has been enhanced by ~ 150 %, when cultured on MCPC for 7 days. The incorporation of 1.5 wt. % of graphene oxide (GO) in hydroxyapatite nano-rod (NRHA) increases the proliferation of MC3T3-E1 cells, by approximately 120 % as compared to control samples, after incubation of 7 days [178]. Fig. 2.11 represents the viability of L929 cells on optimally processed HA- 40 wt. % CaTiO<sub>3</sub> composite [179]. It is clearly observed that the viability of cells significantly increases with increase in incubation period.



Fig. 2.11. The viability of osteoblast SaOS2 cells, culture on control/ (HA-40 wt. % CaTiO<sub>3</sub>; H6C4 )/ (HA-60 wt. % CaTiO<sub>3</sub>; H4C6/(HA-80 wt. % CaTiO<sub>3</sub>; H2C8) samples, after incubation of 3, 5, and 7 days. The symbol (\*) represents the statistically significant difference between the samples, at p < 0.05. The symbol (\*\*) represents the statistically significant significant difference between 3, 5, and 7 days of incubation for each composition, at p < 0.05. [Reproduced with permission from ref (179): Copyright (2010), Wiley Periodicals, Inc.].

#### 2.6.1. Effect of incorporation of piezoelectric secondary phase on cellular response

The piezoelectric biomaterials have been considered as an appealing alternative for orthopedic applications [180]. Piezoelectricity generates surface charge by application of electrical stimulation which favors ontogenesis [181]. It has been suggested that the incorporation of piezoelectric biomaterials as a secondary phase in HA matrix enhances the proliferation of human osteoblast cells [182]. It has been demonstrated that the proliferation of L929 cells on HA- 70 wt. % of BaTiO<sub>3</sub> increases by approximately 201 % as compared to control sample, after incubation of 7days [184]. In another study, the incorporation of 40 wt. % of BaTiO<sub>3</sub> in HA has reported to increase the cellular response

[183]. The density of SaOS2 cells, cultured on HA and HA-90 vol. % BT composite, increased by 87 and 97 % respectively, after incubation of 7 days [184].Overall, the literature reports suggested that the addition of piezoelectric secondary phases in HA enhances the proliferation of osteoblast cells.

#### 2.6.2. Effect of external stimuli on cellular response

The application of external stimuli such as, electric field and surface polarization has been recognized as an alternative technique to enhance the cell growth and proliferation [185]. It has been demonstrated that the optimal electrical stimulation significantly enhances the proliferation and differentiation (osteogenic) of human mesenchymal stem cells on biomaterial surfaces [186]. The adhesion of osteoblast cells on negatively polarized surface increases due to electrostatic interaction between negative charge and  $Ca^{2+}$  ions [187]. Fig. 2.12 demonstrates the effect of electrical polarization on viability of L929 cells, while cultured on HA-BT composites [188]. The negatively polarized HA-x BT (x = 20 and 40 wt. %) composites increases the density of L929 cells by 40 and 62 %, respectively, while cultured on these surfaces for 48 h [187].The viability of osteoblast SaOS2 cells on negatively charged HA-7.5 wt. % ZnO bio-composite has been reported to increase by ~ 77 %, after incubation of 3 days [189].



Fig. 2.12. The viability of L929 cells, cultured on control, HA and HA-40 wt. % BaTiO<sub>3</sub> composites. The symbol (\*) shows the significant difference among the samples, incubated for 5 and 7 days with respect to that incubated for 3 days. The symbols (\*\*) and (\*\*\*) represent the statistically significant differences among the samples with respect to control disk and HA, respectively, at P < 0.05. [Reproduced with permission from ref. (187): Copyright (2014) Wiley Periodicals, Inc].

It has been reported that polarized HA (negatively charged) enhances the proliferation of MC3T3-E1 osteoblast cells, after incubation of 7 days [190]. In another study, it has been reported that the proliferation of human osteoblast like hFOB cells on N -polarized HA is almost double as compared to P-polarized HA, after incubation of 11 days [191]. The proliferation of rBMSCs cells on polarized NKN increases after incubation of 7 days as compared with incubation of 4 days. The polarized NKN (piezoelectric constant ~ 70 pC/N) samples demonstrate higher proliferation of mouse mesenchymal cells as compared to unpolarized samples [192]. The adhesion of osteoblast cells on polarized (22-24 kV/cm) lithium modified NKN (Li.06Na.5K.44NbO3, LNKN) has been increased by approximately 33 % as compared to unpolarized LNKN [194]. It has been observed that the polarization increases the formation of apatite layer in HA by approximately two times as compared to unpolarized HA [193]. Verma et al. [194] demonstrated that

incorporation of sodium potassium niobate (NKN) in 1393 bioglass (1393 BG) enhances the cell proliferation. It has been concluded that the application of external stimuli and surface polarization enhances the proliferation of osteoblast cells.

The adhesion of cells with substrate depends on the interaction of substrate with protein (present in growth media) [195]. The cations such as,  $Ca^+$ ,  $Na^+$ ,  $Mg^+$  and  $K^+$  are adsorbed of on N-polarized surfaces which promotes the formation of bone like apatite layer and increase the cell proliferation [196]. However, anionic groups such as  $HPO_4^{2^-}$  and  $HCO_3^{2^-}$  attracted towards p-polarized surfaces which act as anti-adhesive agents [188]. These anions do not promote the formation of apatite layer. Fig. 2.13 illustrates the interaction of polarized HA surfaces with MC3T3-E1 cells in growth media. It has been suggested that negatively polarized surfaces promote the formation of bone like apatite layer [197].



Fig. 2.13. Schematic diagram, representing the effect of surface polarization on adhesion of MC3T3-E1 cells (a) The inorganic ions, amino acids, and proteins are attached on HA surfaces, (b) normal cell adhesion on uncharged surfaces, positively charged ionic groups such as  $Ca^{2+}$  are attracted towards in negatively charged surface and promote the formation of the bone-like layer on N-polarized HA surface. The positively charged surface actively adsorbs the anions which is not favorable for apatite layer formation and consequently, the cell adhesion. [Reproduce with permission from ref. (197): Copyright (2001), John Wiley & Sons, Inc].

# 2.7. Closure

The effects of addition of various antibacterial elements as secondary phases in different ceramic matrices and application of external stimuli on antibacterial behavior were reviewed. The effect of incorporation of piezoelectric secondary phase on antibacterial response was also reviewed. In addition, the cellular response of tailored biomaterials was discussed. Further, the influence of addition of piezoelectric secondary phases, external stimuli as well as surface polarization on cellular response was also discussed. In addition, the possible mechanisms for antibacterial as well as cellular response were also discussed.

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